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**THE INFLUENCE OF ORIENTED DEFORMATION ON DEEP LEVEL IMPURITIES
AND RADIATION DEFECTS IN SILICON AND ZINC****Nomanjan Sultanov***Doctor
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E-mail: yusupov.fizika@gmail.com***ВЛИЯНИЕ НАПРАВЛЕННОЙ ДЕФОРМАЦИИ НА ГЛУБОКИЕ ПРИМЕСИ
И РАДИАЦИОННЫЕ ДЕФЕКТЫ В КРЕМНИИ И ЦИНК****Султанов Номанжон Акрамович***д-р техн. наук,
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The tensosensitivity of deep levels of radiation defects (RD) and impurity atoms has been investigated by DLTS method. The high sensitivity (100 - 150 meV/GPa) has been found for level of RD ($E_c - 0,33$ eV) to orientational deformation. The energy of ionization of the others levels investigated depends on pressure weakly. The tensosensitivity of levels of E- and K-centers is increases after annealing, but symmetry and others parametrs of centers do not change.

АННОТАЦИЯ

Методом DLTS исследована тензочувствительность глубоких уровней радиационных дефектов (РД) и примесных атомов. Обнаружена высокая чувствительность (100 - 150 мэВ/ГПа) уровня RD ($E_c - 0,33$ эВ) к ориентационной деформации. Энергия ионизации остальных исследованных уровней слабо зависит от давления. Тензочувствительность уровней E- и K-центров после отжига увеличивается, но симметрия и другие параметры центров не изменяются.

Keywords: tensosensitivity, defects, spectroscopy, impurity atoms, oriented strain, annealing, A-centre, center symmetry.

Ключевые слова: тензочувствительность, дефекты, спектроскопия, примесные атомы, ориентированная деформация, отжиг, А-центр, симметрия центра.

Introduction. Determination of the atomic state of an impurity or intrinsic defect that forms deep levels (DL) in the forbidden band is of great interest for the physics of local states in semiconductors. A decrease in the lattice symmetry under uniaxial deformation, in the presence of defects located asymmetrically in the lattice, can lead to anisotropic change in their parameters, which does not coincide with the anisotropy of the semiconductor, and to the splitting of multiply degenerate energy levels [2;4]. The symmetry of the defect can be determined from the results of measuring the shift in the energy position E_a of the DL and its splitting under uniaxial deformation.

Oriented deformation is used to study the optical, electrical and paramagnetic properties of semiconductors and local centers. In recent years, non-stationary capacitive spectroscopy - NCS DL - has found wide application for the study of centers with DL. This method has a high sensitivity and makes it possible to study low concentrations of DL centers, including non-paramagnetic centers and nonradiative recombination centers. However, the NCS DL method, combined with

oriented deformation, has been used for the study of DL relatively recently [2]. In this work, the effect of pressure P on the properties of the A center in silicon was measured. The effect of uniaxial deformation on the parameters of other DL in silicon has been relatively poorly studied. The dependence of E_a on the value of hydrostatic pressure has been studied in more detail. Table 1 shows the literature data on the results of measuring the average coefficient of tensosensitivity $\beta = E_a/\Delta P$ for some centers with PG [3,4,5].

The aim of this work was a preliminary study of the influence of oriented deformation on the parameters of DLs formed by radiation defects (RD) and impurity atoms in silicon.

Samples and methods of measurement. For the preparation of samples, silicon of the SEP-5 and SHB-10 grades with a resistivity $\rho = 5$ and 10 Ohm.cm , respectively, was used. NCS DL measurements carried out with the help of deposited Schottky barriers showed that in the initial silicon the total concentration of DL did not exceed $\sim 10^{11} \text{ cm}^{-3}$ (Fig. 1, curve 1)

Table 1.

Impurity level

№	Impurity	Level, eV	β , meV / GPa	Literature
1	S	$E_c - 0,59$	-20	/4/
		$E_c - 0,32$	-17	
2	Se	$E_c - 0,52$	-21	/6/
		$E_c - 0,30$	-18	
3	Te	$E_c - 0,37$	-12	/5/
		$E_c - 0,19$	-9	
4	Au	$E_v + 0,35$	0,5	/3/
		$E_c - 0,54$	-15	
5	Pt	$E_c - 0,235$	-24	/6/
		$E_c - 0,49$	-28	
		$E_v + 0,32$	-14	

Silicon alloying with metals was carried out by diffusion at temperatures of $1000 - 1200 \text{ }^\circ\text{C}$ for 4-8 hours, followed by rapid cooling (quenching). Before diffusion, the crystal surface was thoroughly cleaned [6]. After diffusion, a $50 - 100 \text{ }\mu\text{m}$ thick layer with a high diffusant content was removed from the surface of the samples and the plates were cut into samples with dimensions of $1 \times 1 \times 6$ or $1.5 \times 1.5 \times 6 \text{ mm}^3$.

The long sides of the samples were oriented along the (111), (110) and (100) axes. On one of the sides of the samples, Schottky barriers were created by depositing Au on n-Si or Sb on p-Si. In some cases, p - n junctions were used that were created by diffusion prior to doping of silicon with centers with DL.

Samples for studying the RD had the same dimensions and orientation. Both Schottky barriers and diffusion p ~ n junctions were used as rectifying contacts. Si was irradiated with γ -quanta of the ^{60}Co isotope at 300

K and a source intensity of $3,4 \cdot 10^{12} \text{ kV} / \text{cm}^2 \cdot \text{c}$. The irradiation dose Φ did not usually exceed $1 \cdot 10^{18} \text{ kV} / \text{cm}^2$, because, at a higher Φ , the base was compensated and it became too high resistance for measuring the NCS DL [8].

The NCS DL spectra were measured in the constant-capacitance mode using the setup described in [6]. Its absolute sensitivity is 10^2 pF , the ratio of the voltage recording times $t_2/t_1=3$, the gating windows varied within $0.5 - 500 \text{ ms}$, the diode voltage usually did not exceed 10 V .

Typical NCS DL spectra of n- and p-Si after irradiation and annealing are shown in Fig. 1 and 2. The parameters of some RDs are given in Table 2 and are in good agreement with the literature data [5-7].

Uniaxial deformation was created using a lever system, which in its design was almost similar to that described in [2]. The system used made it possible to create

a pressure of up to 4 GPa with a sample area of 1 mm². To measure the NCSDL spectra, the sample together with the holder was cooled to the temperature of liquid nitrogen. The sample was heated to 300 K rather slowly - in about an hour. The temperature T was measured with a copper - constantan thermocouple.

The sensitivity of most of the investigated DL to uniaxial deformations turned out to be relatively low, and to determine the effect of pressure P on the ionization energy E_a of the level, measurements were carried out in the switching mode: with a slow change in T, the pressure was changed from P = 0 to the maximum value.

In fig. 3 shows a typical NCSDL peak measured in this mode. The change in E_a was estimated from the temperature shift of the NCSDL peak:

$$\Delta E_a = E_a(0) - E_a(P) = E_a(0) \left(\frac{T_1}{T_2} - 1 \right) = E_a(0) \frac{\Delta T}{T}$$

where E_a (0) and E_a (P), T₁ and T₂ are the ionization energies of the DL and the temperature of the maxima of the peaks at P = 0 and P > 0, respectively.

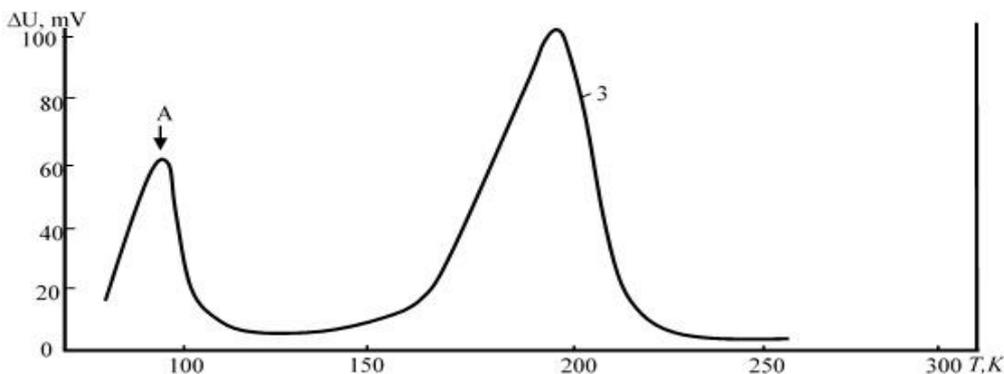


Figure 1. Spectra of NCSDL n-Si before irradiation (curve 1), after irradiation up to $\Phi = 1 \cdot 10^{18} \text{ cm}^{-2}$ (2) and after heat treatment at 250° for 20 minutes (3)

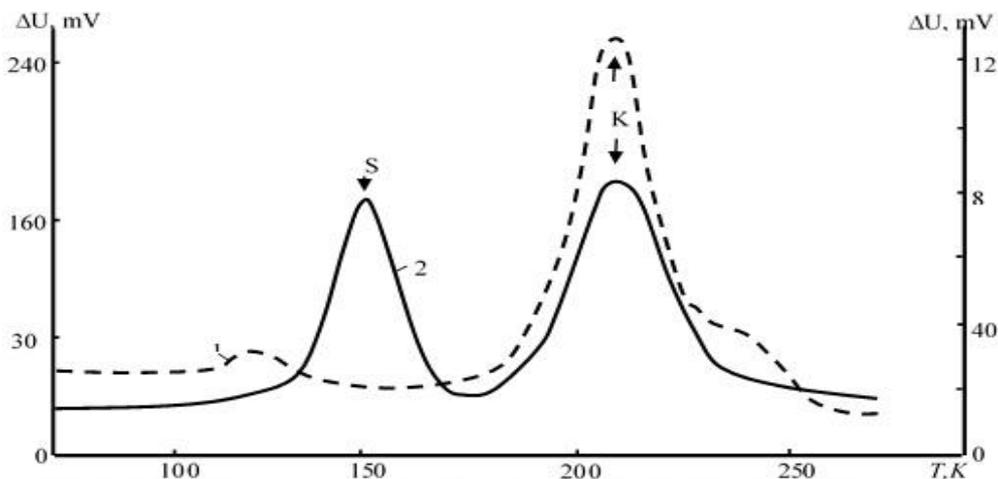


Figure 2. NESGU spectrum of irradiated p-Si ($\Phi = 1 \cdot 10^{18} \text{ cm}^{-2}$) before (curve 1) and after (curve 2) annealing at 250° C for 20 minutes

If the DL had a high sensitivity to pressure and there was a noticeable shift of the NCSDL peaks at P > 0, then the level parameters were determined in the usual way

by changing the gating windows. The accuracy of determining ΔE_a in this case was of the order of $\pm 0.02 \text{ eV}$.

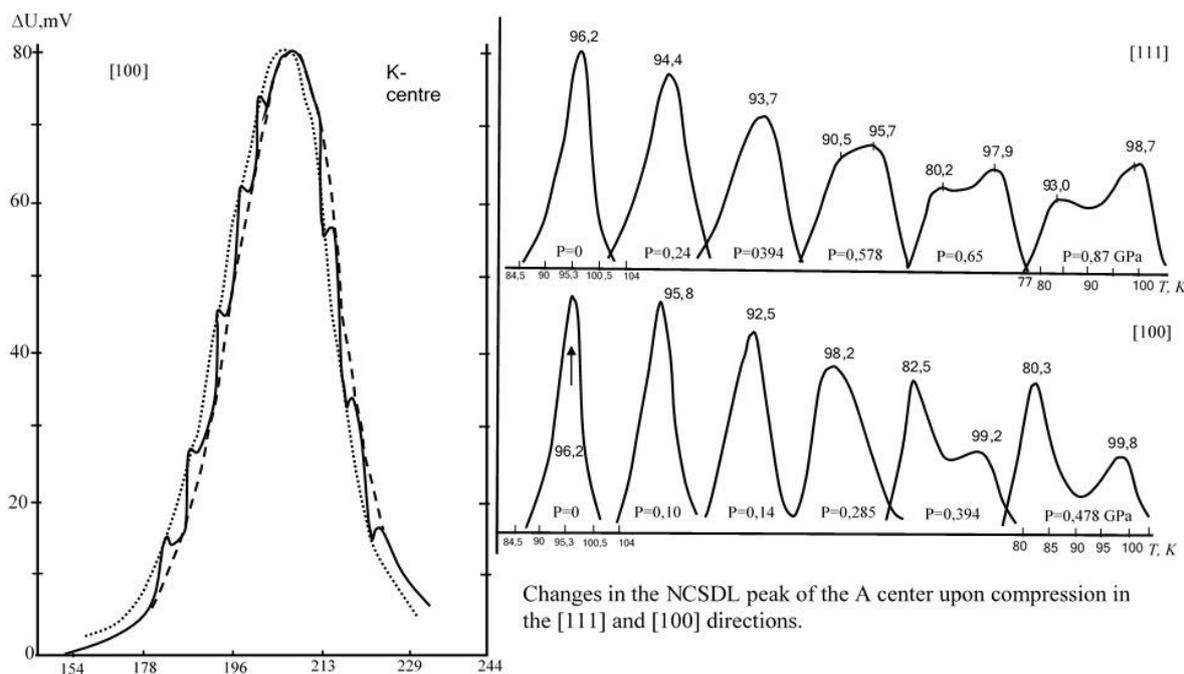


Figure 3. Typical view of the NCSDL peak (K ~ center before HT) with the commutation method of measurement. Curve 1 – experimental dependence, 2,3 – interpolated curves at P=0.3 GPa and P = 0, respectively

Measurement results and discussion. Radiation defects in n-Si. Immediately after irradiation in p-Si, a series of RDs with DLs are formed (Fig. I, curve 2). The tensosensitivity of the A- and E-centers was investigated in detail. The rest of the RDs had a low concentration and for them E_a and the electron capture cross sections σ_n were determined only at P = 0: $E_a = 0.23, 0.3, 0.37,$ and 0.53 eV from the bottom of the conduction band and, $\sigma_n = 2 \cdot 10^{-16}, 4 \cdot 10^{-15}, 4 \cdot 10^{-16}$ and $3 \cdot 10^{-13}$ cm², respectively.

A-center. After irradiation, the A center with $E_a = E_c - 0.17$ eV dominates in n-Si diodes, because the original silicon was obtained by the Czochralski method and had an oxygen concentration of $N_{to} \approx 2 \cdot 10^{17}$ cm⁻³. The A-center had a high sensitivity to mono pressure. In fig. 4 shows the dependence of the NCSDL peak for this center on P in the (100) and (111) directions, and in Table. 2 - β values obtained in this work and in work [2]. The results of both works are in good agreement with each other.

L-center. Annealing the irradiated samples at 250 °C for 20 minutes leads to a decrease in the concentration of all RDs. In particular, the concentration of the A-center decreases from $1,5 \cdot 10^{14}$ to $1 \cdot 10^{13}$ cm⁻³.

As a result of annealing, a new RD is formed, called the L-center, with a concentration of the order of $1 \cdot 10^{13}$ cm⁻³. The width of the NCSDL peak associated with this center corresponds to a DL with a fixed ionization energy and at P=0 $E_a = E_c - 0,33$ eV. The electron capture cross section, determined from the Arrhenius line, is of the order of $4,6 \cdot 10^{-16}$ cm². The nature of the L-center is debatable and, possibly, it is a divacancy-oxygen complex [7]. With further annealing, the concentration of the L-center decreases and the field of heat treatment (HT) at 350°C for 2 hours, its concentration becomes below the sensitivity limit of the setup.

The L center has a high tensosensitivity, and at P> 0.4 GPa, the NCSDL peaks are split in all crystallographic directions. Figure 5 shows, as an example, the dependence of the NCSDL peak in the (111) direction on P. More details of the dependence of the NCSDL peaks on P are given in [7]. Comparison with the theory [1] shows that, apparently, the L - center has rhombic symmetry.

Radiation defects in p-Si. Immediately after p-Si irradiation, several RD levels appear in the lower half of the band gap (Fig. 2, curve I). The tensosensitivity of the K-center was investigated in detail. The amplitudes of the other peaks were small to determine β with sufficient accuracy, and for these levels E_a and the capture cross section of holes σ_n were determined at P = 0: $E_a = 0.2$ and 0.48 eV from the top of the valence band, $\sigma_n = 7 \cdot 10^{-16}$ and $2 \cdot 10^{-13}$, respectively.

Impurity levels. Zinc forms in silicon two acceptor states with $E_a = E_v + 0,28$ and $E_v + 0,62$ eV, which are attributed to one zinc atom at the sites, as well as a DL with $E_a = E_c - 0.47$ eV, which is associated with zinc atoms at interstices [10-13].

Upon deformation, the NCSDL peaks of all zinc levels shift to lower T, but these displacements are small. The isotropic change in E_a of the acceptor states $E_v + 0.28$ and $E_v + 0.62$ eV confirms the assumption that zinc atoms associated with these DL are located in the sites of the silicon lattice. Small values of β show that zinc at the sites behaves like a shallow acceptor impurity despite the large absolute value of E_a .

Platinum. With diffusion alloying, platinum forms in silicon acceptor $E_c - 0.26$ eV and donor $E_v + 0.36$ eV levels [9], as well as level $E_c - 0.53$ eV. In p-Si, acceptors prevail, in p-Si - donors. Under uniaxial compression,

the NCSDL peaks of all levels are shifted insignificantly. In silicon doped with vanadium, chromium, nickel, and sulfur, the effect of uniaxial compression on E_a was investigated only in the (111) direction.

Conclusion. Thus, of the investigated DL created by RDs and impurity atoms, only the levels of the A and L centers have a noticeable tensosensitivity and their

splitting occurs under the action of oriented deformation. E_a of other investigated DL depends much weaker on P. It is of interest to change the tensosensitivity of E and K-centers after low-temperature treatment. In this case, neither the parameters of the center (the NCSDL peak does not shift along the temperature axis) nor its symmetry change.

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